

## Preparation of WO<sub>3</sub> Nano-material Negative Electrode for Asymmetric Supercapacitor

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**Abstract.** Tungsten oxide nanomaterials were deposited on flexible conductive carbon fiber sheets by a simple thermal evaporation method and compared with conventional hydrothermal tungsten oxide nanomaterials. The morphology and composition of the product were characterized by X-ray diffraction and scanning electron microscope. The electrochemical properties of WO<sub>3</sub> nano-electrode materials were tested by three-electrode system. The results showed that the area capacitance of WO<sub>3</sub> nanoelectrode was 0.24F/cm<sup>2</sup> at a current density 10mA/cm<sup>2</sup>, which was similar to that of tungsten oxide composite electrode prepared by complex evaporation plating process. After charging and discharging, the capacitance retention rate can be maintained about 80%, proving that WO<sub>3</sub> nanocrystalline has large capacitance, good redox reversibility and electrochemical stability, and can be used as negative electrode material for asymmetric supercapacitors.

### Introduction

Energy storage devices with high efficiency have given rise to lots of attention because of more and more urgent demand for environmental conservation and coping with the increasingly serious depletion of fossil fuels [1]. The rapid development in high-power electronic devices, emergency power supplies and hybrid electric vehicles provide the potential application for these energy storage devices [2, 3]. Among them, the advent of supercapacitors (SCs) as a new class of energy storage appeal for a large amount of scientists due to their advantages of high-power density, excellent reversibility, fast recharge ability, and long cycle life[4,5]. However, encountered with the problem of low energy density, the further application of supercapacitors is under restrictions. On account of the equation of energy density,  $E = 0.5 CV^2$ , to increase the energy density of SCs, it is essential to augment the output voltage and/or the capacitance [6]. It is effective to increase the voltage up to 3V using organic and ionic liquid electrolytes, such as lithium perchlorate in propylene carbonate or tetraethyl ammonium tetrafluoroborate in acetonitrile, but the enhancement of voltage is at the high cost, poor ionic conductivity, and high toxicity [7].

An alternative to increasing voltages is developing asymmetric supercapacitors (ASCs) with a battery type Faradaic positive electrode as energy source and a capacitive negative electrode as a power source, together with environmentally friendly aqueous electrolytes which have higher ionic conductivities. Different from symmetric supercapacitors (SSCs), by combining the voltage windows of two different electrodes, ASCs are able to increase the maximum operation voltage up to 2V in aqueous electrolyte, and thus power and energy density of the device.

For all great progress made in improving the capacitance of positive electrode materials, the research on negative electrode materials is seldom carried out [8, 9]. Even of Carbon-based nanomaterials are commonly used as negative electrode in virtue of their high surface area, excellent electrical conductivity, and large power density [7], their low specific capacitance weakens the further enhancement of the energy density for ASCs. Therefore, it is desired to search novel negative electrode materials for ASCs.

Some negative electrodes based on metal oxide such as Co<sub>3</sub>O<sub>4</sub>/RuO<sub>2</sub>, [10,11] MoO<sub>3</sub>·x, [12], V<sub>6</sub>O<sub>13</sub> [13] and iron oxide [14] have shown much higher energy density than carbon-based

materials; However, they suffered from the low electronic conductivity, which has a negative impact on their electrochemical performance. In order to address this problem, the selection of high-specific-area and high-conductivity nanomaterials, such as tungsten oxide ( $\text{WO}_{3-x}$ ) nanowires, [15, 16] to load electrochemically active materials is worth considering. However, the research on using tungsten oxide for supercapacitors is relatively few because it is hard to grow this kind of nanomaterials by conventional hydrothermal method due to its high melting points.

In this communication, we reported a different method from hydrothermal synthesis, which is depositing tungsten oxide ( $\text{WO}_3$ ) by evaporation. In addition, the performance of the negative electrode materials using the two methods will be compared. The capacitance performance of  $\text{WO}_3$  nanomaterial as negative electrode is improved markedly when deposited, achieving areal capacitance of  $0.24 \text{ F/cm}^2$  at current density of  $10 \text{ mA/cm}^2$  and capacitance retention of 80% after 10000 cycles at a scan rate of  $200 \text{ mV/s}$ .

## Experimental

All reagents used in the experiment are up to the standard of analytical grade that were used without any further purification. The  $\text{WO}_3$  samples were fabricated on a carbon cloth substrate by using hydrothermal method and evaporation method separately. First group, 10 mmol of  $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$ , 30 mmol of  $\text{NH}_4\text{F}$ , and 25 mmol of  $\text{CO}(\text{NH}_2)_2$  were dissolved in 60 mL of DI water under stirring, respectively. After 10 minutes of slight stirring, two Teflon-lined stainless-steel autoclaves of 25 mL capacity were filled with 20 mL of homogeneous solution prepared above respectively. Then, carbon clothes ( $2 \text{ cm} \times 3 \text{ cm}$ ) were cleaned by ethanol, deionized water, and using ultrasonic treatment in turn and then immersed into the reaction solution. Subsequently, the two autoclaves were sealed and heated at  $120^\circ\text{C}$  for 12 hs (Sample 1) and  $150^\circ\text{C}$  for 12hs (Sample 2) and then allowed to cool down to room temperature. Second group, 20 mmol of  $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$ , 20 mmol of  $\text{Na}_2\text{SO}_4$ , and were dissolved in 40 mL of DI water under stirring, respectively. The autoclave was sealed and heated at  $160^\circ\text{C}$  for 24 hs (Sample 3). Third group, 20 mmol of  $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$  and 20 mmol of  $\text{NaCl}$  were dissolved in 40 mL of DI water under stirring, respectively adjusting the solution to  $\text{PH}=2$  with  $\text{HCL}$  solution. The autoclave was sealed and heated at  $160^\circ\text{C}$  for 24 hs (Sample 4). The other process of Sample 3 and 4 are the same as Sample 1 and 2.

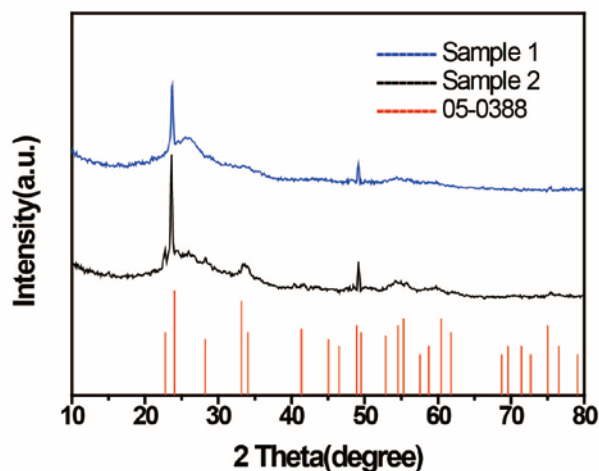
Two cleaned carbon clothes used as substrate were cleaned by ethanol, deionized water, the 0.65g of ultrafine tungsten powder was placed onto the bottom of tungsten boat evenly and then ethanol was dripped into the powder to avoid the powder dropping out when the boat is upside down. Put the boat and the carbon clothes to their corresponding position in evaporation coating machine. The flow rate is that Argon flow 60sccm and Oxygen flow 0.5SCCM. After 20 minutes of venting, Evaporation power starts to heat the boat with the rate of  $1\text{A/s}$ . The temperature rose to  $1120^\circ\text{C}$ , remaining for 15 minutes and then stop heating. The sample E was taken out after the 15 minutes of cooling.

## Material Characterization and Electrochemical Measurement

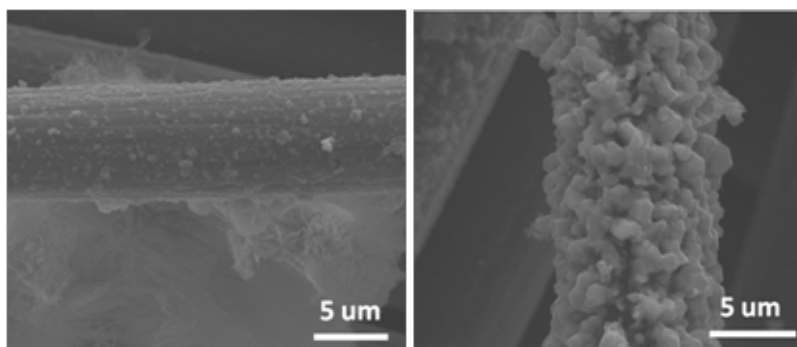
The composition morphology and structure of the electrode materials were characterized by X-ray diffraction patterns, and Field emission SEM (FE-SEM, JSM-6330F). CV and galvanostatic charge/discharge measurements were conducted using an electrochemical workstation (CHI 660D). The electrochemical measurements were performed in a three-electrode cell, with the individual sample used as the working electrode, a Carbon rod as the counter electrode and a SCE as the reference electrode.

## Results and Discussion

Figure 1 shows the X-ray diffraction (XRD) patterns of the sample 1 and sample5, indicating that the peaks of both samples. Can be indexed to the standard card (JCPDS Card No. 05-0388), which are in accordance with Tungsten Oxide ( $\text{WO}_3$ ).



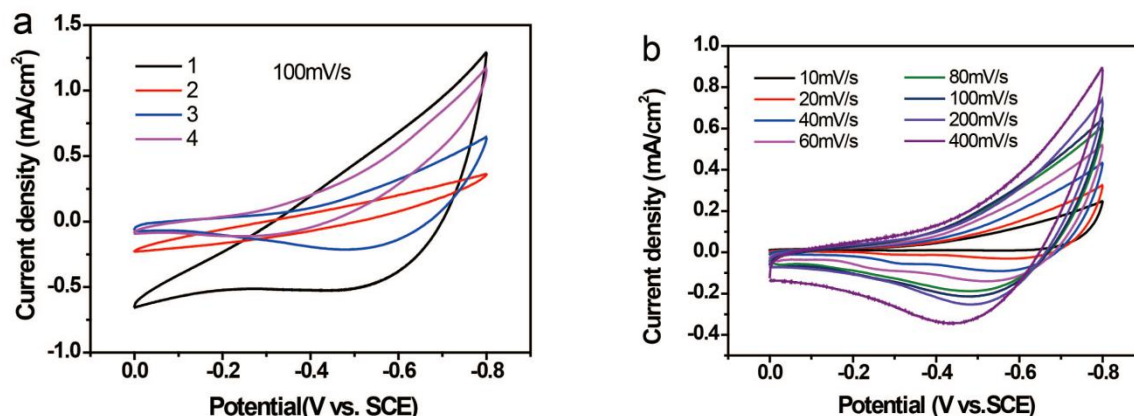
**Fig. 1** XRD spectra collected for  $\text{WO}_3$  electrodes.



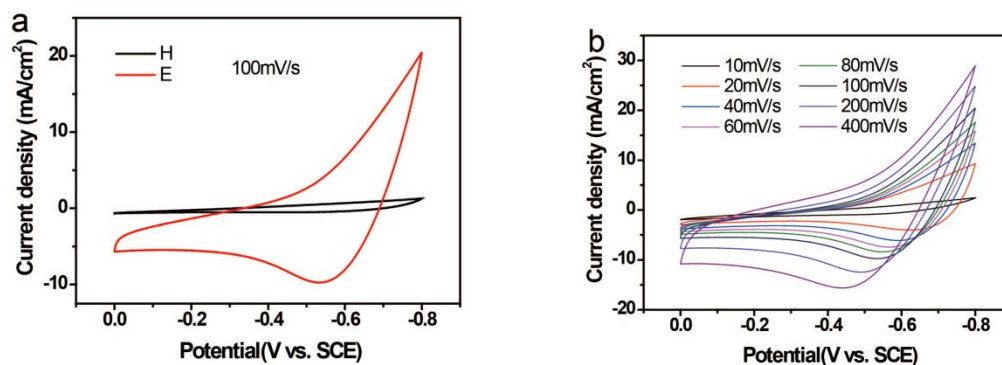
**Fig. 2** SEM images of (a)  $\text{WO}_3$  nanomaterials by hydrothermal, (b)  $\text{WO}_3$  nanomaterials by evaporation on a carbon cloth

The SEM images of  $\text{WO}_3$  on the carbon clothes of Sample1 and Sample2 are shown in Figures 2(a), (b) respectively. It can be observed from Figure 1a that the carbon fibers were dotted with not a few  $\text{WO}_3$  nano-particles on Sample 1 by hydrothermal method, with a diameter about 1nm. In contrast, it can be seen in Figure 2(b) that the distribution of  $\text{WO}_3$  nano-particles in Sample5 are more uniform than that in Sample1, with a diameter about 2nm and the number much more than that of  $\text{WO}_3$  nano-particles in Sample1, indicating that evaporation deposition is more beneficial to the growth of  $\text{WO}_3$  nano-particals than hydrothermal method. As a result, the specific surface area got increased on electrode materials, combined with the high conductivity of carbon fiber, and the capacitance of the negative electrode is, therefore, augmented.

The electrochemical properties of the  $\text{WO}_3$  electrodes were investigated in a three-electrode cell in 5 M LiCl aqueous electrolyte, with a Carbon rod counter-electrode and a SCE reference electrode. Cyclic voltammogram (CV) curves of  $\text{WO}_3$  electrodes for Sample 1, 2, 3 and 4 at a scan rate of 100 mV/s were compared in Figure 3(a). It is clear that the area of the CV curve for Sampe 1 is larger than that for other three samples, and furthermore, the CV profiles of the Sample 1 at various scan rates remain similar as the scan rate increases from 10 to 200 mV/s (Figure 3(b)), indicating that hydrothermal process used for Sample1 is more suitable than that for other samples.



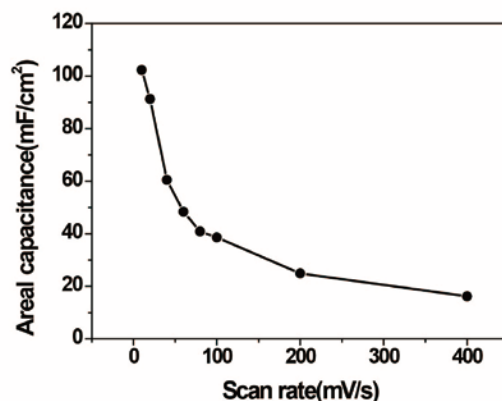
**Fig. 3** (a) CV curves collected for Sample 1, 2, 3 and 4 negative electrodes at a scan rate of 100 mV/s. (b) CV curves collected for Sample 1 negative electrode at various scan rates.



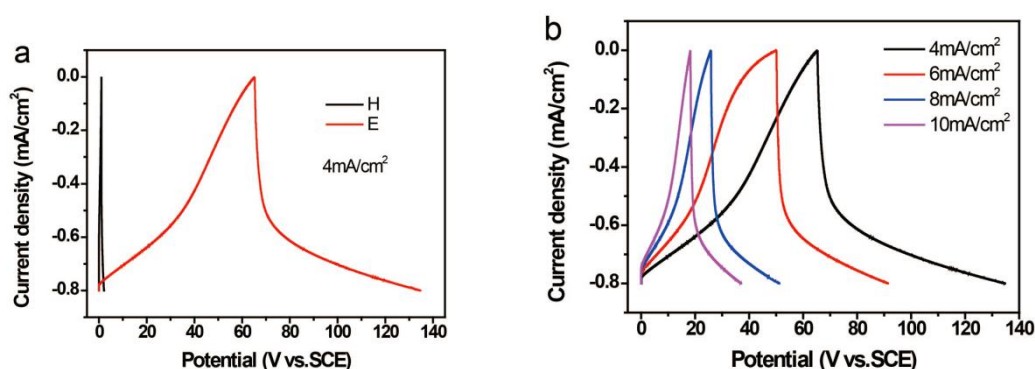
**Fig. 4** (a) CV curves collected for WO<sub>3</sub> negative electrodes by both hydrothermal(H) and evaporation(E) method at a scan rate of 100 mV/s. (b) CV curves collected for WO<sub>3</sub> negative electrodes by evaporation(E) method at various scan rates.

Cyclic voltammetry (CV) curves of Sample1 (noted as H) and Sample5 (noted as E) electrodes at a scan rate of 100 mV/s are compared in Fig. 4a. As is shown, the current density of the evaporation Sample5 electrode is substantially higher than that of the Hydrothermal Sample1 electrode, indicating that the substantial enhancement of electrochemical capacitance originated from the a mass of growth of WO<sub>3</sub> nano-particles by evaporation process. Moreover, it can be observed in Fig.4b that the CV curves of the Sample5 at various scan rates remain similar as the scan rate increases from 10 to 400 mV/s, making it clear the significantly enhanced capacitive properties and high-rate capability of the WO<sub>3</sub> electrode by evaporation process.

Figure 5 illustrates the areal capacitances of the Sample5 at the scan rates from 10 mV/s to 400 mV/s calculated on the basis of its CV curves, which is the same order of magnitudes as what was reported about negative electrode for ASCs in Advanced Energy Materials. The fabrication of WO<sub>3</sub> electrode is much simpler than that of WO<sub>3-x</sub>/MoO<sub>3-x</sub> electrode.



**Fig. 5** Areal capacitance of the  $\text{WO}_3$  negative electrode as a function of the scan rate.

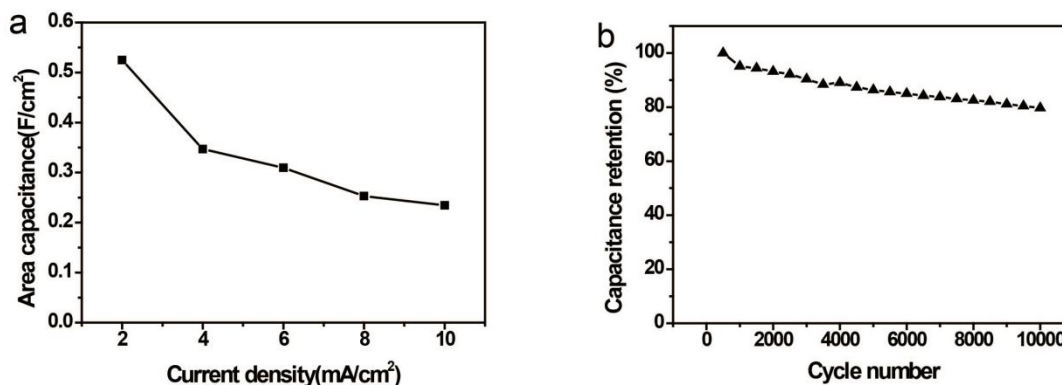


**Fig. 6** (a) Galvanostatic charge-discharge curves of the  $\text{WO}_3$  negative electrodes by both hydrothermal (H) and evaporation (E) method at  $4 \text{ mA/cm}^2$ . (b) Galvanostatic charge-discharge curves of the  $\text{WO}_3$  negative electrodes by evaporation collected at different current densities.

The galvanostatic charge-discharge curves of the Sample1 and Sample5 electrodes at a current density of  $4 \text{ mA/cm}^2$  are shown in Figure 6(a). Compared with Sample1, Sample 5 exhibits slower and more symmetrical charge and discharge curves and a more linear variation of potential vs. time, which is also supported by the galvanostatic charge-discharge curves collected for Sample5 electrode at different current densities from 2 to 3, 4, 5, 6, 8, and  $10 \text{ mA/cm}^2$  (Figure 6(b)), proving the greatly improved capacitance, Columbic efficiency, and reversibility of the  $\text{WO}_3$  electrode at a high current density by evaporation process.

The areal capacitance of  $0.24 \text{ F/cm}^2$  at current density of  $10 \text{ mA/cm}^2$  (shown in Figure 7(a)) can be up to  $\text{WO}_{3-x}/\text{MoO}_{3-x}$  electrode (about  $0.24 \text{ F/cm}^2$ ), the best value for  $\text{WO}_3$ . Figure 7(b) shows the capacitance retention of Sample5 as a function of cycle number. The capacitance retention of  $\text{WO}_3$  is still about 80% of the original capacitance after 10000 cycles at a scan rate of  $200 \text{ mV/s}$ , indicating that  $\text{WO}_3$  by evaporation is of excellent long cycle life, demonstrating that  $\text{WO}_3$  by the evaporation process in this paper is a suitable candidate for negative electrode.





**Fig. 7** (a) Areal capacitance of the WO<sub>3</sub> negative electrodes by evaporation collected from galvanostatic charge-discharge curves as a function of current density. (b) Cycling performance of the WO<sub>3</sub> negative electrodes by evaporation at 200 mV/s for 10000 cycles

## Conclusion

In conclusion, a feasible method has been explored to grow WO<sub>3</sub> nanomaterials on carbon clothes by evaporation. The electrochemical performances of WO<sub>3</sub> negative electrode have been significantly improved by changing the fabrication process from conventional hydrothermal method to evaporation method. Moreover, stable rate capability and cycle capability were also achieved. These encouraging performances have access to its applications in high-performance asymmetric super-capacitors, opening up a novel approach to advancing the performance of energy storage devices. Moreover, stable rate capability and cycle capability were also achieved. These encouraging performances have access to its applications in high-performance asymmetric super-capacitors, opening up a novel approach to advancing the performance of energy storage devices.

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