

## Two-Step Fabrication of Porous $\alpha\text{-Fe}_2\text{O}_3\text{@TiO}_2$ core-shell Nanostructures with Enhanced Photocatalytic Activity

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**Abstract.** A new synthetic approach has been developed to prepare  $\alpha\text{-Fe}_2\text{O}_3\text{@TiO}_2$  core-shell nanostructures at ambient conditions (e.g., in water,  $\leq 100^\circ\text{C}$ ). This approach shows a few unique features, including: short reaction time (a few minutes) for forming core-shell nanostructures, no requirement of high temperature calcinations for generating  $\text{TiO}_2$  (e.g., at  $80\text{--}100^\circ\text{C}$ ), tunable  $\text{TiO}_2$  shell thickness, high yield and good reproducibility. The experimental results show that the  $\alpha\text{-Fe}_2\text{O}_3\text{@TiO}_2$  core-shell nanostructures exhibit enhanced photocatalytic activity compared to the pure  $\text{TiO}_2$  (P25) and pure  $\text{Fe}_2\text{O}_3$  in degradation of organic dye molecules with visible light irradiation. This could be attributed to the large surface area of  $\text{TiO}_2$  nanoparticles for maximum adsorption of dye molecules, and the effective charge separation at the heterojunction of  $\alpha\text{-Fe}_2\text{O}_3$  and  $\text{TiO}_2$ . The findings may open a new strategy to synthesize  $\text{TiO}_2$ -based photocatalysts with enhanced efficiency for environmental remediation applications.

### Introduction

The design and fabrication of core-shell nanoparticles have attracted much more attention because of their unique shape, size and structure-dependent properties [1, 2]. The synergetic effects between the core and shell materials may drastically improve the overall performance of the hybrid composites, which results in a range of potential applications, including biomedical application [3, 4], catalysis [5, 6], electronics [7] and chemical sensors [8]. The properties of core-shell nanostructures can be adjusted by changing either the constituting materials or the core to shell ratio [9].

Among the well-known semiconductors, titanium oxide ( $\text{TiO}_2$ ) nanoparticles have been regarded as one of the most promising nanomaterials because of its abundance, low-cost, long-term stability and nontoxicity [10]. In particular,  $\text{TiO}_2$  coating on different metals or metal oxide nanoparticles as  $\text{TiO}_2$ -based core-shell nanocomposites have shown high potential research interests [11]. However, the photocatalytic efficiency of  $\text{TiO}_2$  is restricted by a few factors such as the fast electron-hole recombination and the limited utilization of sun light in visible light range (400–800 nm). Above issues impede its broad applications in energy and environmental areas. Preparation of heterojunctions of core@ $\text{TiO}_2$  may induce the transition of the optical response from the UV to visible region, and separate the electrons and the holes into different regions of the catalyst [12, 13]. Alpha-iron oxide ( $\alpha\text{-Fe}_2\text{O}_3$ ) is a promising material for catalysis because of its stability, favourable band gap, low cost and abundance [14–16]. The combination of  $\text{TiO}_2$  with  $\text{Fe}_2\text{O}_3$  to form a core-shell structured nanocomposite can decrease the recombination rate of photogenerated electrons and holes. Moreover, the core-shell structured  $\text{Fe}_2\text{O}_3\text{@TiO}_2$  nano-composites can be used to improve the visible light response of  $\text{TiO}_2$  [17, 18].

In this study, we demonstrate a facile coating method to fabricate porous 1D  $\alpha\text{-Fe}_2\text{O}_3\text{@TiO}_2$  core-shell structures by control the condensation of tetrabutyl titanate (TBT) in acetone solution for constructing uniform porous  $\text{TiO}_2$  shells. This method is very simple, yet important, which can be extended for coating  $\text{TiO}_2$  shells on different cores and allows varying the thickness of  $\text{TiO}_2$  shells. In addition, no high temperature calcination is needed for crystallization into anatase phase. Moreover,

the as-prepared 1D  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@TiO<sub>2</sub> core-shell structures exhibits highly enhanced photo degradation performance under visible light irradiation.

## Experimental

### Preparation of $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> Particles

In a typical procedure, 6.758 g of FeCl<sub>3</sub>·6H<sub>2</sub>O was dissolved in 250 mL of distilled water to prepare a 0.1 M FeCl<sub>3</sub>·6H<sub>2</sub>O solution. This solution was subsequently heated at 80°C for 16 h to obtain akaganeite ( $\beta$ -FeOOH) nanorods. These nanorods were then collected by centrifugation and thoroughly washed with deionized water and 1.0 M NaOH solution several times to remove excess chlorine (Cl) ions and naturally dried at 60°C for 5 h. Upon drying, the  $\beta$ -FeOOH nanorods were calcined in air at 400 °C for 2 h to convert them to porous  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanorods.

### Synthesis of $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@TiO<sub>2</sub> core-shell nanoparticles

In a typical procedure, 0.05 mL titanium (IV) butoxide 97% (TBT) was added to 10 mL ethylene glycolate (EG). The mixture was magnetically stirred for 8 h at room temperature, noted as Solution A. 3 mL  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> colloidal suspension (0.01 M) was poured into 10 mL acetone under stirring for around 5 min, noted as Solution B. Then, 0.5 mL Solution A was added in Solution B, and left it for ageing 1 h. The precipitates (titanium precursor) were harvested by centrifugation and then dispersed in boiling water for bath ~2 h. The white grey colloids were centrifuged at 3,000 rps for 10 min, then washed with alcohol, and finally dried in vacuum at 60 °C for a few hours to generate  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@TiO<sub>2</sub> core-shell particles.

### Characterizations

The shape and microstructure was confirmed by TEM (JEOL 1400), operated at an accelerated voltage of 100 kV. The TEM specimen was prepared by dropping solution onto the copper grid and dried in air naturally. The crystalline structure of the composite colloids was further characterized by high resolution TEM (HRTEM), Philips CM200, at an accelerating voltage of 200 kV. To observe surface, SEM (Hitachi 900) was used and operated at 20 kV. To further understand the surface, XPS analysis was conducted with a Physical Electronics PHI 5000 Versa probe spectrometer with Al K $\alpha$  radiation (1486 eV). Analysis of the spectra was done using the Physical Electronics Multipak software package. The composition analysis of final products was conducted by XRD (Philip MPD diffractometer) with Cu K $\alpha$  radiation with scanning step of  $2\theta = 0.02^\circ/s$ . The UV-visible absorption spectrum was obtained on a Shimadzu UV-2600 UV-Vis spectrophotometer (Varian) with a 1-cm quartz cell. The surface area of the as-synthesized powders and pore size distribution were measured by BET equipment (TriStar 3000) via nitrogen gas adsorption and desorption isotherms.

### Photocatalysis Test

The photocatalytic activities of TiO<sub>2</sub> based core-shell nanocomposites were determined by measuring the decolourization of methylene blue (MB) under simulated solar light irradiation in a batch reaction. The reactor volume is 250 mL. A 300W xenon lamp (PLS-SXE300C) equipped with AM 1.5G total reflection filters was used to obtain simulated solar light to trigger the photocatalytic reaction, the lamp was positioned about 15 cm away from the reactant solution meniscus. The luminance of the light source over the reactant solution was 0.7 W/cm<sup>2</sup>. A 100 mL solution of 30 ppm MB was injected into the reaction system, while 0.020 g of the photocatalyst was added. Prior to irradiation, the suspension was magnetically stirred in the dark for 30 min to ensure adsorption-desorption equilibrium. At given time intervals, 3 mL solution was sampled and filtrated to remove the catalysts. The filtrates were analysed by recording the variations of the absorption band maximum (657nm) of MB using a Shimadzu UV-2600 UV-Vis spectrophotometer. After various time intervals, MB concentration could be estimated using the following equation:

$$\text{MB concentration} = C/C_0 \times 100\% \quad (1)$$

Where  $C_0$  is the initial MB absorbance at 657 nm and  $C$  is the absorbance obtained after various intervals of time. To eliminate the effect of heating during irradiation, the reactor was equipped with reflux condensation. After decolourization, the supernatant of the solution was obtained via centrifugation, and then characterized by UV-Vis spectroscopy.

## Experimental

### Morphology and Composition

As revealed by SEM and TEM images in fig. 1A and B, uniform spindle-shaped  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> spindle-shaped nanorods were first synthesized with a diameter of  $\sim 100$  nm and length of  $\sim 400$  nm as the cores using hydrothermal method. Then, via a sol-gel process and condensation of titanium glycolate in acetone, uniform core-shell  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@TG spindle-shaped structures can be obtained. After water boiling in air, the size of spindle-shaped structures are retained the same, however, as shown in the SEM and TEM image (fig. 1C and D), the surface of the shell become rough. The XRD pattern of the water boiled sample (Fig. 2A) exhibits the characteristic diffraction peaks, which can be well indexed to anatase TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. The XRD pattern of the achieved nanocomposites contains the peaks mainly attribute to  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (JCPDS no.33-0644), a few peaks attribute to TiO<sub>2</sub>. This may be caused by the very small weight fraction of the TiO<sub>2</sub> shell, compared with that of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> cores in the whole core-shell nanocomposites. No other peaks related to impurities are observed, which indicates the high purity of the product. XPS analysis was used to obtain further information regarding the surface structure and composition of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@TiO<sub>2</sub> core-shell nanostructures. Fig. 2B shows the characteristic energy peaks of the elemental titanium, oxygen and iron, with binding energies of Eb=459.08 eV (Ti2p), Eb=530.64eV (O1s) and Eb=710.68 eV (Fe2p), respectively. The C1s peak originated from the adventitious carbon. The XPS survey confirms that the TiO<sub>2</sub> shell exists on the surface of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> particles.

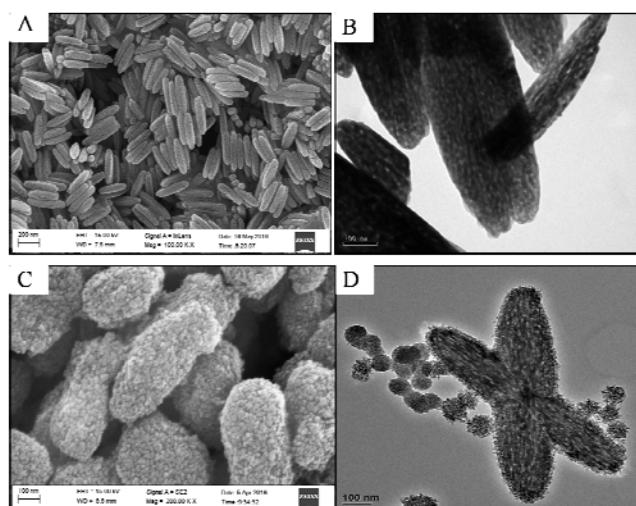


Fig. 1 (A) SEM image of as-prepared  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanorods; (B) TEM image of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanorods; (C) SEM image of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanorods coated with TiO<sub>2</sub> nanoparticles; (D) HRTEM image of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@TiO<sub>2</sub> core-shell nanostructure.

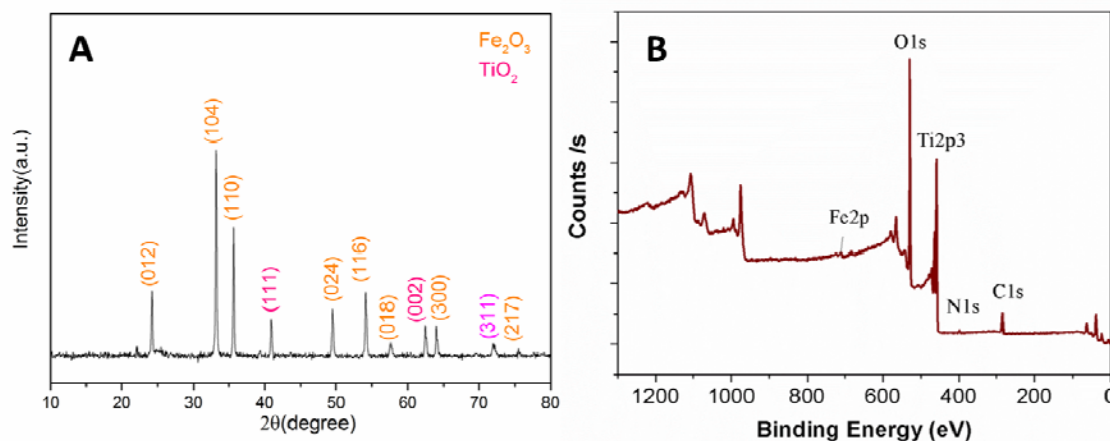


Fig. 2(A) XRD pattern of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@TiO<sub>2</sub> core-shell nanostructures; and (B) XPS spectra of the as-prepared  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@TiO<sub>2</sub> core-shell nanostructures.

### Optical Property

Fig. 3 shows the UV-Vis spectra of pure TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@TiO<sub>2</sub> core-shell nanostructures. For pure TiO<sub>2</sub> particles, the absorption peak appeared at  $\sim$ 320 nm. However, when  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanorods are coated with TiO<sub>2</sub> nanoparticles, a wide absorption peak appears in the range of 400 nm to 800 nm. Comparing with pure TiO<sub>2</sub> particles, it was found that there is an obvious red shift in the absorption peak from UV region to visible light region, which is highly beneficial to enhancing the photocurrent performance and photocatalytic activity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@TiO<sub>2</sub> core-shell nanostructures under solar light irradiation.

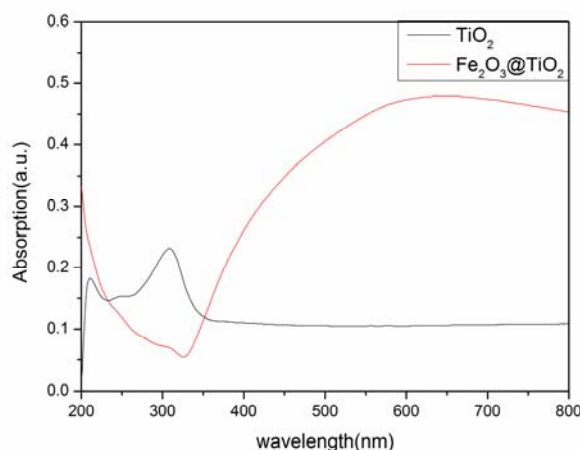


Fig. 3 UV-vis spectra of TiO<sub>2</sub> nanoparticles and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@TiO<sub>2</sub> core-shell nanostructures.

### BET Study

The Brunauer-Emmett-Teller (BET) surface area and pore volume of the as-prepared  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@TiO<sub>2</sub> nanocomposites are calculated to be 152 m<sup>2</sup>/g and 5.8 cm<sup>3</sup>/g, respectively. After coating with TiO<sub>2</sub> shells, the BET surface area is greatly increased (The surface area and pore volume of pure  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is 52.5 m<sup>2</sup>/g and 8.6 cm<sup>3</sup>/g). This results from the densification of TiO<sub>2</sub> networks and the growth of nanocrystals. The corresponding pore size distribution curves (Fig. 4B) derived from the adsorption branches of the isotherms by using the Barrett-Joyner-Halenda (BJH) method clearly show a smaller pore size than that of the uncoated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanorods, further illustrating those small amorphous nanoparticles are aggregated, and grow into large anatase nanocrystals. Such large mesoporesites

derived from the uniform  $\text{TiO}_2$  nanoparticles, in association with the very thin layer of  $\text{TiO}_2$ , are expected to show high performances when they are applied to photocatalytic reactions.

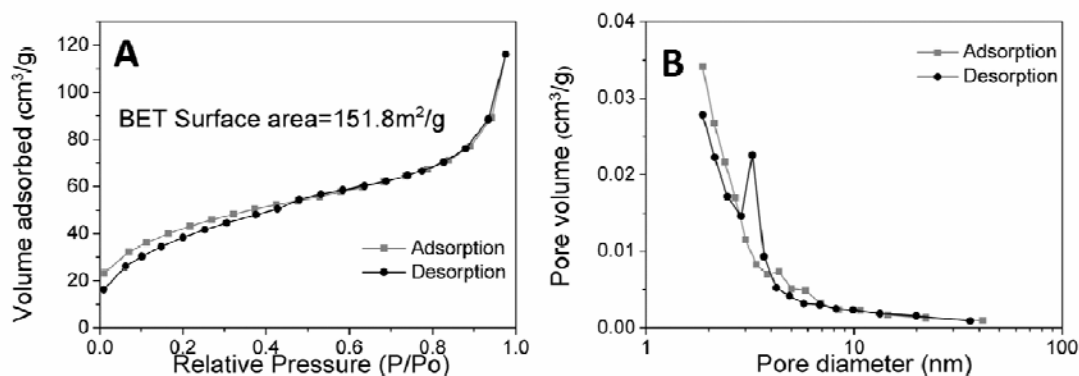


Fig. 4 BET measurements for  $\alpha\text{-Fe}_2\text{O}_3 @\text{TiO}_2$  core-shell nanostructures (A) and (B) with a surface area of  $S_{\alpha\text{-Fe}_2\text{O}_3} = 151.8 \text{ m}^2/\text{g}$  and an average pore size of 5.8 nm.

### Photocatalytic Study

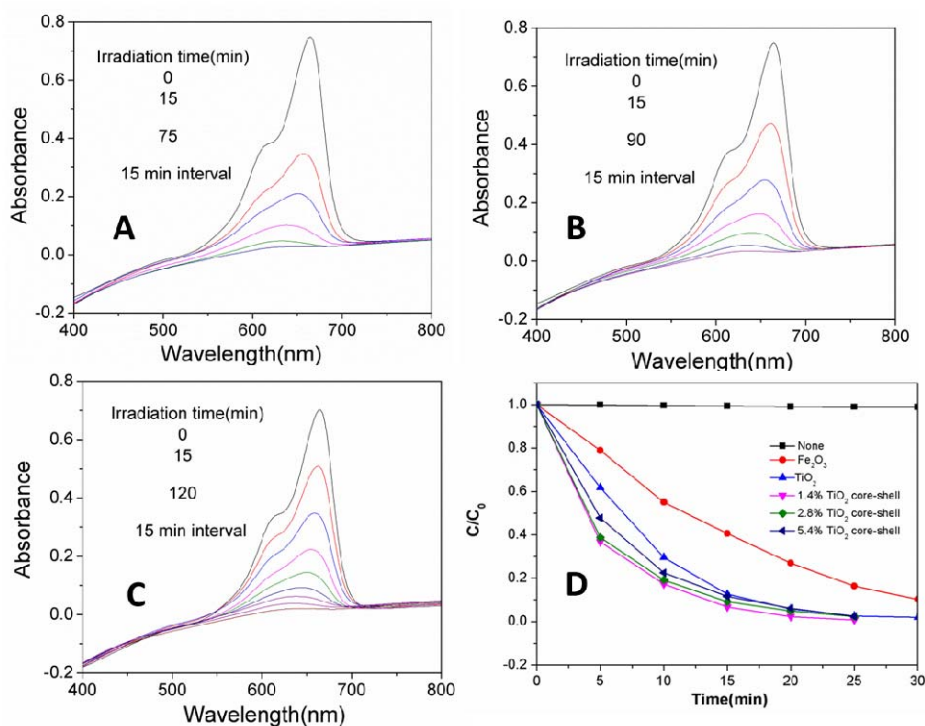


Fig. 5 Absorbance changes of MB molecule during visible light irradiation with different photocatalyst (A)  $\alpha\text{-Fe}_2\text{O}_3 @\text{TiO}_2$  core-shell nanocomposites with (A) 1.4% wt. (B) 2.8% wt. and (C) 5.4% wt.  $\text{TiO}_2$  coating; (D) Concentration change of MB in the presence of pure  $\text{TiO}_2$  nanoparticles, pure  $\alpha\text{-Fe}_2\text{O}_3$  nanorods and  $\alpha\text{-Fe}_2\text{O}_3 @\text{TiO}_2$  core-shell nanostructures with different  $\text{TiO}_2$  coating amount under visible light irradiation.

The photocatalytic performance of the  $\alpha\text{-Fe}_2\text{O}_3 @\text{TiO}_2$  core-shell structures was evaluated by monitoring the degradation of methylene blue (MB) in aqueous solution under UV and visible light irradiation. The organic dye suspension with the photocatalyst was kept in dark conditions for 30 minutes to ensure the establishment of adsorption-desorption equilibrium of MB on the sample surfaces. Fig. 5A-C show the absorbance changes of MB molecular during visible light irradiation with photocatalysts of  $\alpha\text{-Fe}_2\text{O}_3 @\text{TiO}_2$  core-shell nanocomposites with (A) 1.4% wt. (B) 2.8% wt. and



(C) 5.4% wt.  $\text{TiO}_2$  coating. As can be seen that the core-shell structure with 1.4% wt.  $\text{TiO}_2$  coating shows the fastest photocatalytic efficiency, the dye molecular can be totally degraded with 75 min (Figure 5A). Followed with that of 2.8 % wt.  $\text{TiO}_2$  coating, nearly 100% degradation can be achieved with 90 mins at the same conditions (Fig. 5B). While the core-shell structure with 5.4% wt.  $\text{TiO}_2$  coating shows the slowest degradation rate, fully photo-decomposition happens after 120 mins (Fig.5C). Fig. 5D shows the corresponding concentration changes of the MB solution as a function of visible light exposure time, with the as-prepared pure  $\alpha\text{-Fe}_2\text{O}_3$ ,  $\text{TiO}_2$  and  $\alpha\text{-Fe}_2\text{O}_3@ \text{TiO}_2$  core-shell structures with different amount of  $\text{TiO}_2$  coating as photocatalysts.

MB solution alone without photocatalyst was conducted to test the self-photo degradation of MB. It can be seen that the photocatalytic activity of the  $\alpha\text{-Fe}_2\text{O}_3@ \text{TiO}_2$  core-shell structures can be tuned via controlling the coating amount of  $\text{TiO}_2$  shells. The thinner the coating is, the faster the visible light photocatalytic activities can be achieved. The visible light photocatalytic performance of all the  $\alpha\text{-Fe}_2\text{O}_3@ \text{TiO}_2$  core-shell nanocomposites is superior to that of pure  $\text{TiO}_2$  and  $\alpha\text{-Fe}_2\text{O}_3$ .

Under visible light irradiation,  $\text{TiO}_2$  is unable to be excited to generate electron-hole pairs, whereas  $\text{Fe}_2\text{O}_3$  could be activated and yield charge carriers[19]. Subsequently, the photo generated electrons immigrate from the conduction band of  $\text{Fe}_2\text{O}_3$  to the conduction band of  $\text{TiO}_2$  under the action of built-in electric field and the concentration gradient activated and yielded charge carriers, while photogenerated holes accumulated in the valence band of  $\text{Fe}_2\text{O}_3$ . The negative electrons in the conduction band of  $\text{TiO}_2$  further react with molecular oxygen  $\text{O}_2$  dissolved in the solution to form the superoxide anion  $\text{O}_2^-$  and hydrogen peroxide  $\text{H}_2\text{O}_2$ . While the accumulated holes in the valence band of  $\text{Fe}_2\text{O}_3$  react with  $\text{OH}^-$  species existing on the surface of the catalyst, producing reactive hydroxyl radicals ( $\cdot\text{OH}$ ), which is the predominant species in the process of photocatalysis .

## Conclusion

In summary, the  $\text{Fe}_2\text{O}_3@ \text{TiO}_2$  core-shell nanostructures have been fabricated via a simple but efficient synthesis route. This synthesis strategy shows a few advantages in generating anatase  $\text{TiO}_2$  shells, such as rapid and effective surface coating, no need of high temperatures ( $>500^\circ\text{C}$ ) treatment. The as-prepared  $\text{Fe}_2\text{O}_3@ \text{TiO}_2$  core-shell nanocomposites display a large surface area and high stability. In particular,  $\text{Fe}_2\text{O}_3@ \text{TiO}_2$  composites have been proved to show superior visible light photocatalytic activity to the pure  $\text{TiO}_2$  nanoparticles and  $\text{Fe}_2\text{O}_3$  nanorods. The narrow bandgap of  $\text{Fe}_2\text{O}_3$  was employed to extend the optical response of  $\text{TiO}_2$  to visible light. It is shown that the  $\text{Fe}_2\text{O}_3@ \text{TiO}_2$  heterojunction structures favours the charge transfer and suppress rapid recombination of photo-induced electrons and holes.

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