Effect of Single SiO₂ Layer Incorporation on Electrical Performances of Metal-insulator-metal Capacitors with Al₂O₃-HfO₂-Al₂O₃ Dielectrics

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Abstract. The metal-insulator-metal (MIM) capacitors with Al₂O₃/HfO₂/Al₂O₃ (AHA) and Al₂O₃/HfO₂/SiO₂/HfO₂/Al₂O₃ (AHSOA) dielectric structure using atomic layer deposition (ALD) technique have been fabricated. It is demonstrated that the dielectric loss at higher applied frequency and quadratic VCC in high- κ MIM capacitors can be decreased by means of introducing an ultrathin layer of ALD SiO₂, which has an opposite sign of quadratic VCC against the HfO₂ layers. A high capacitance density of 8.1 fF/ μm^2 and the low leakage current density of 2.5×10^{-7} A/cm² at 1 MV/cm can be achieved with the dielectric structure as AHSOA. In addition, the breakdown electrical field in AHSOA dielectric is obviously improved. Furthermore, with the survey of different leakage models, the Schottky emission has been considered as the dominating conduction mechanism for both of the AHA and AHSOA capacitors.

Introduction

High-permittivity (high- κ) metal-insulator-metal (MIM) capacitors have recently been studied for RF and analog/mixed-signal integrated circuits [1-7]. Among high- κ materials, HfO₂ is one of the most promising candidates due to high- κ (about 20-25) and large band gap (5.7 eV) [8-10]. However, a relatively high quadratic voltage coefficients of capacitance (α) remains as a serious dilemma for the application of the HfO₂ MIM capacitors, where the α is known to be inversely proportional to dielectric thickness and dielectric constant [11-13]. So that, the α and capacitance density have a trade-off relationship. Meanwhile, as for conventional dielectric materials of MIM capacitors, SiO₂ and Si₃N₄ are commonly used materials although it is hard to achieve high capacitance density due to their low dielectric constant κ (about 4-7) [4, 14, 15]. Therefore, in order to obtain high capacitance density, relatively low α and low leakage current density, it seems to be a good choice to incorporate single SiO₂ or Si₃N₄ layer into high- κ sandwiched dielectric materials of MIM capacitors.

In addition, due to the requirement of the MIM capacitors embedded into the inter-level dielectric layers, the maximum fabrication temperature of the MIM capacitors should be compatible with the thermal budget (<450 °C) of Cu back-end-of-line process. Of the various high- κ dielectric deposition techniques, atomic layer deposition (ALD) technique, which has some outstanding advantages such as exquisite control over the film composition and structure, high uniformity and perfect conformity, can satisfy the low temperature requirement. So far, many reports have been focused on high- κ deposition with ALD, where well properties of high capacitance density, small voltage coefficient of capacitance and low leakage current were observed [16, 17]. However, there has been little study on the effect of single SiO₂ in high- κ sandwiched dielectrics deposited by ALD

in advanced MIM capacitors.

In this paper, we show that the α of $\text{Al}_2\text{O}_3/\text{HfO}_2/\text{Al}_2\text{O}_3$ dielectric MIM capacitance can be decreased by introducing an ultrathin SiO_2 layer deposited by ALD, which has an opposite sign of the α . The ALD Al_2O_3 layer is deposited as the contact layer between dielectric and electrodes in order to obtain the low leakage current density of the MIM capacitors. Meanwhile, different current leakage models have been used to fit the experimental data in order to investigate the dominating conduction mechanism.

Experimental

A 150 nm TiN was grown on the 500 nm thermal oxide/Si substrate by sputtering, served as the bottom electrode. Subsequently, the TiN coated wafers were transferred into the ALD chamber for the dielectric growth. Al_2O_3 and HfO_2 were grown with thermal ALD process from $\text{Al}(\text{CH}_3)_3/\text{H}_2\text{O}$ and $\text{Hf}(\text{N}(\text{CH}_3)(\text{C}_2\text{H}_5))_4/\text{H}_2\text{O}$, respectively. The single SiO_2 layer was deposited by plasma enhanced ALD process from the precursor of Trisdimethylamino-silane (TDMAS: $\text{SiH}(\text{N}(\text{CH}_3)_2)_3$) and O_2 , with the plasma power of 100 W. During the process, the reactor chamber temperature was set at 200°C , and all of the vapors of precursors were delivered into the reactor chamber with N_2 gas as carrier. The dielectric thickness was measured by an ellipsometer. Finally, the top electrode with 100 μm diameter of sputtered Ta was formed with lift-off method. The MIM capacitors with two structures of dielectric layers [A (1nm)/H(11nm)/A(1nm), A(1nm)/H(5nm)/S(1nm)/H(5nm)/A(1nm), A, H and S denoted as Al_2O_3 , HfO_2 , and SiO_2 , respectively] were fabricated to investigate the effect of the intermediate SiO_2 layer. And then we denoted them as AHA and AHSHA capacitors, respectively. Capacitance-voltage (C-V) measurements were carried out with LCR meter (Keithley 4200) with 100-mV ac sweeping signal. The leakage current characteristics were measured by using semiconductor device analyzer (Agilent B1500A).

Results and Discussion

Fig. 1 (a) Schematic diagram of the AHA MIM capacitor; (b) Schematic diagram of the AHSMA MIM capacitor; (c) Cumulative distributions of capacitance densities of both MIM capacitors

Fig. 1 shows the schematic diagrams of AHA and AHSMA MIM capacitors and the cumulative distributions of the measured capacitance densities at 0 V under 1 MHz for both capacitors. The resulting capacitance density equals 12.6 and 8.1 fF/μm² for the AHA and AHSMA capacitors at 50% cumulative probability, respectively. It can be found that with the introduction of the SiO₂ layer, the capacitance density decreases as expected. The dielectric constants can be calculated as 18.4 and 11.9 according to the capacitance density for the AHA and AHSMA capacitors, separately. For the AHA capacitor, the dielectric constant is in accordance with other reports [4, 11, 18]; while with the introduction of single SiO₂ layer, the dielectric constant drops. The frequency dependences of capacitance densities for both capacitors at 0 V dc bias is shown in Fig. 2, together with the dissipation factor. It can be observed that the capacitance densities are relatively stable as the measurement frequency varied from 1 KHz to 1 MHz, indicating their excellent dielectric characteristics with frequency. Meanwhile, the variation of the dissipation factor along with frequency (as shown in Fig. 2) implies AHSMA structure capacitor with 1 nm intermediate SiO₂ layer can effectively reduce dielectric loss relative to AHA structure capacitor at higher applied frequency.

Fig. 2 The capacitance densities as a function of frequency for both capacitors, together with the dissipation factor.

The voltage coefficients of capacitance (VCC) are very important parameters for the application of MIM capacitors to radio frequency circuits, which can be determined by using Eq. (1) to fit the experimental data:

$$C(V) = C_0(\alpha V^2 + \beta V + 1), \quad (1)$$

Where C_0 is the measured capacitance at 0 V, α and β represent quadratic and linear VCC, respectively.

Fig. 3 (a). Typical plotting of C/C_0 versus voltage at 1 MHz for both capacitors, together with the fitted curves; (b). Quadratic V_{cc} as a function of frequency for both capacitors.

The typical normalized capacitances (C/C_0) versus dc bias measured at 1 MHz for both capacitors together with the fitted curves have been shown in Fig. 3(a). It is found that the α value is 4000 ppm/V² for the AHA capacitor; while the α is decreased to 2090 ppm/V² evidently with the introduction of the 1nm SiO₂ layer in AHSMA capacitor, as it is predicted. The asymmetric of the curves for dc bias maybe result from the oxidation of the bottom TiN electrode, which leads to the relatively large VCC due to electron injection from bottom electrode under positive bias voltage [16-18]. At the same time, a better VCC characteristic in AHSMA capacitor can be obviously found, but a worse one in AHA capacitor under positive bias voltage. Fig. 3(b) exhibits the α as a function of frequency for both capacitors. For the AHA capacitor, the α value varies from 5340 to 4000 ppm/V² with the measurement frequency increasing from 1 KHz to 1 MHz. And for the AHSMA capacitor, the α value varies from 3090 to 2090 ppm/V² with the measurement frequency increasing from 1 KHz to 1 MHz. This results clearly demonstrate the α value is strongly affected by the frequency and decreasing with frequency increase, in agreement with other reports [4, 16].

The typical leakage current density (J) of the capacitors as a function of electric field (E) measured at room temperature is shown in Fig. 4. It is observed that the AHSMA capacitor exists the much higher breakdown field than that of the AHA capacitor. This indicates that the intermediate SiO₂ layer is very effective for improving the dielectric breakdown strength, because SiO₂ has an amorphous state and a large band gap (9 eV) in a universal IC process temperature range [14]. Ding et al. have reported that the crystallization of HfO₂ can be enhanced with increasing its thickness [8]. So with the introduction of the intermediate SiO₂ layer, the crystallization of HfO₂ can be suppressed and the grain boundaries (highly leaky paths) can be decreased, in order to lead to high electrical breakdown strength [19]. Meanwhile, the decrease of leakage current density is found with the introduction of SiO₂ layer, where the leakage current density is 8.1×10^{-7} A/cm² at 1 MV/cm for the AHA capacitor and the leakage current density of the capacitor decreases to 2.5×10^{-7} A/cm² at 1MV/cm with the introduction of the 1 nm intermediate SiO₂ in AHSMA capacitor, when it maintains the fixed physics thickness of the insulator. Furthermore, the small asymmetry of the leakage current density has been found as a function of the electric field for both capacitors, which may be caused by the different work function of the

electrodes.

Fig. 4 Dependence of leakage current density on electric field for both capacitors.

In order to investigate the intrinsic physics of the leakage current, we survey the general leakage mechanisms at present including Schottky emission and Poole-Frenkel (PF) emission [4, 14]. According to Schottky emission arising from a difference in Fermi level between a metal and a semiconductor, charges have to overcome the potential barrier created by the energy difference between the metal and semiconductor [4, 20, 21]. The current density (J_s) can be expressed as

$$J_s = AT^2 \exp\left(-\frac{\phi - \sqrt{q^3 E / 4\pi\epsilon_0\epsilon_r}}{k_B T}\right), \quad (2)$$

where A is the Richardson constant, T is the temperature, ϕ is the height of the Schottky barrier, q is the electronic charge, ϵ_0 is the permittivity of free space, ϵ_r is the dynamic relative dielectric constant, and k_B is the Boltzmann constant. For Poole-Frenkel (PF) emission, it involves the thermal ionization of trapped carriers into the conduction of thin films [14, 21], and can be expressed as

$$J_{PF} = BE \exp\left(-\frac{E_t - \sqrt{q^3 E / \pi\epsilon_0\epsilon_r}}{k_B T}\right), \quad (3)$$

Where B is the constant, E_t is the trap ionization energy. Fig. 5 exhibits the typical plotting of $\ln(J)$ vs $E^{1/2}$ for the AHA and AHSMA capacitors under electron bottom injection according to the Schottky emission. It is observed that the experimental data can be well linearly fitting with

Schottky emission model and the relative dielectric constants (ϵ_r) for AHA and AHSMA dielectrics are deduced around 26.8 and 24.5, which are slightly larger than the previous calculated dielectric constant 18.4 and 11.9 by C-V measurement; while both of the deduced relative dielectric constants for AHA and AHSMA are over 30 with PF emission model (not shown here), which are far larger than values of theory and C-V measurement [8-10]. This result demonstrates the Schottky emission dominates the conduction for both of AHA and AHSMA capacitors, in agreement with the observation in AHA capacitor by Ding et al [4].

Fig. 5 Typical plotting of $\ln(J)$ vs $E^{1/2}$ for the AHA and AHSMA capacitors according to the Schottky emission, and the deduced relative dielectric constants (ϵ_r).

Conclusions

The $\text{Al}_2\text{O}_3/\text{HfO}_2/\text{Al}_2\text{O}_3$ dielectric MIM capacitors with the SiO_2 intermediate layer have been fabricated with ALD technology. The decrease of the quadratic voltage coefficient of HfO_2 MIM capacitance by introducing an ultrathin SiO_2 layer has been realized in our experiment. Meanwhile, it is found that the properties of the dielectric loss and leakage density can be obviously improved. Moreover, the Schottky emission is responsible for the dominating conduction mechanism for both of AHA and AHSMA MIM capacitors.

Acknowledgments

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